

## 1 Selective Modification of Halloysite Lumen with 2 Octadecylphosphonic Acid: New Inorganic Tubular Micelle

3 Weng On Yah,<sup>§</sup> Atsushi Takahara,<sup>\*,§,†,||</sup> and Yuri M. Lvov<sup>\*,‡</sup>

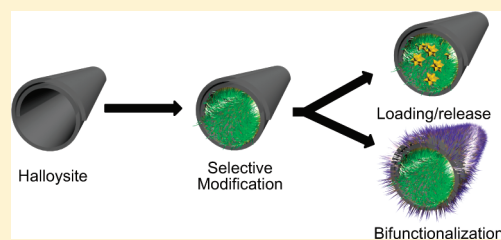
4 <sup>§</sup>Graduate School of Engineering and <sup>†</sup>Institute of Materials Chemistry and Engineering, Kyushu University, 744 Motooka, Nishi-ku,  
5 Fukuoka 819-0395, Japan

6 <sup>||</sup>International Institute for Carbon-Neutral Energy Research (WPI-I<sup>2</sup>CNER), Kyushu University, 744 Motooka, Nishi-ku, Fukuoka  
7 819-0395, Japan

8 <sup>‡</sup>Institute for Micromanufacturing, Louisiana Tech University, 911 Hergot Avenue, Ruston, Louisiana 71272, United States

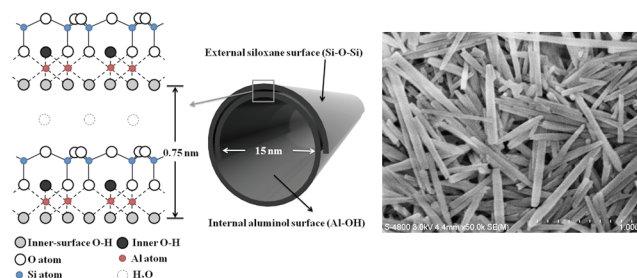
9 **S** Supporting Information

10 **ABSTRACT:** Selective fatty acid hydrophobization of the inner surface of  
11 tubule halloysite clay is demonstrated. Aqueous phosphonic acid was found to  
12 bind to alumina sites at the tube lumen and did not bind the tube's outer  
13 siloxane surface. The bonding was characterized with solid-state nuclear  
14 magnetic resonance (<sup>29</sup>Si, <sup>13</sup>C, <sup>31</sup>P NMR), Fourier transform infrared (FTIR),  
15 and X-ray photoelectron spectroscopy. NMR and FTIR spectroscopy of  
16 selectively modified tubes proved binding of octadecylphosphonic acid within  
17 the halloysite lumen through bidentate and tridentate P–O–Al linkage.  
18 Selective modification of the halloysite clay lumen creates an inorganic  
19 micelle-like architecture with a hydrophobic aliphatic chain core and a hydrophilic silicate shell. An enhanced capacity for  
20 adsorption of the modified halloysite toward hydrophobic derivatives of ferrocene was shown. This demonstrates that the  
21 different inner and outer surface chemistry of clay nanotubes can be used for selective modification, enabling different  
22 applications from water purification to drug immobilization and controlled release.



### 23 ■ INTRODUCTION

24 Tubule halloysite clay has garnered interest in material science  
25 due to its versatile features of large surface area, high porosity,  
26 and tunable surface chemistry which enabled this nanomaterial  
27 to be utilized as a catalyst,<sup>1</sup> in electronic devices,<sup>2</sup> for  
28 entrapment of hydrophilic and lipophilic active agents,<sup>3</sup> and  
29 as a nanofiller for polymers.<sup>4</sup> Halloysite comprises naturally  
30 occurring aluminosilicate nanotubes with a 1:1 Al:Si ratio and a  
31 stoichiometry of  $\text{Al}_2\text{Si}_2\text{O}_5(\text{OH})_4 \cdot n\text{H}_2\text{O}$ . Halloysite occurs  
32 mainly in two polymorphs: the anhydrous form, with and  
33 interlayer of 7 Å, and the hydrated form, with expanded  
34 interlayer spacing of 10 Å, as a result of the incorporation of  
35 water in the interlamellar space.<sup>4d–f</sup> Halloysite tubes consist of  
36 gibbsite octahedral sheet (Al–OH) groups on the internal  
37 surface and siloxane groups (Si–O–Si) on the external surface  
38 as shown in Figure 1.<sup>5</sup> This difference results in a negatively  
39 charged outer surface and a positively charged inner lumen in  
40 the pH range 2–8. Interestingly, halloysite nanotubes have  
41 surface chemistry opposite to that of imogolite.<sup>6</sup>  
42 Addition of 4–5% clay nanotubes into plastics typically  
43 doubles the composite strength and halloysite capacity to  
44 encase and release chemically active agents in a sustained  
45 manner, promising new functional nanocomposite materials.  
46 Recent studies have shown that halloysite is a biocompatible  
47 material; thus it is attractive for biotechnology, pharmaceutical,  
48 and medical research.<sup>7</sup> The medical applications of these clay  
49 nanotubes include cancer cell separation, bone implants,  
50 cosmetics, and controlled drug delivery.<sup>8</sup> Halloysite biocompo-



**Figure 1.** (Left) Schematic illustration of crystalline structure and (right) FE-SEM image of halloysite nanotubes.

51 sites<sup>9</sup> are perspective materials with the great advantages that  
52 they are derived from natural resources, environmentally  
53 friendly, and available at low cost.<sup>22</sup>

54 Selective modification with functional molecules for different  
55 surfaces offers great promise for organic/inorganic composites  
56 (for example, simultaneous selective modification of gold–  
57 metal oxide systems based on high affinity of thiols for the gold  
58 surface and of carboxylic acid for the oxide areas).<sup>10</sup> Sequential  
59 bifunctionalization of Al–SiO<sub>2</sub>/Si surfaces first by thiols and  
60 then by organosilanes and area-selective binding of organo-  
61 silanes to oxidized areas of Si(111)–H surfaces have also been  
62 reported.<sup>11</sup> Recently, silane hydrophobization of single-wall

**Received:** November 12, 2011

63 imogolite nanotube lumen was demonstrated.<sup>11a</sup> A possibility  
64 to produce halloysite tubes with hydrophobized lumen will  
65 widen their applications as nanocontainers for nonpolar  
66 chemicals and for water purification.

67 Halloysite nanotubes having aluminum innermost and  
68 silicate outermost surfaces allow for different inner/outer  
69 surface chemistry. Selective modification between silica and  
70 alumina in halloysite is difficult. Attempts to use organosilanes  
71 for aluminosilicate surface modification resulted in silane  
72 binding to both inner and outer surfaces.<sup>13</sup> Organophosphorus  
73 compounds are more promising for selective binding because  
74 they react selectively with TiO<sub>2</sub> areas of prepatterned TiO<sub>2</sub>/  
75 SiO<sub>2</sub> substrates.<sup>12</sup> Due to the high affinity of organophosphorus  
76 molecules toward metal oxide surfaces, aluminosilicate clays are  
77 expected to present similar selectivity. The use of organo-  
78 phosphonic acids in selective modification is attractive for  
79 preparation of highly hydrolytic stable hybrid materials, the  
80 lumen modification of which can be performed in water at  
81 ambient conditions, in contrast to organosilane coupling agents.  
82 Inorganic micelle-like tubes with inner aliphatic chains and  
83 outer hydrophilic silicate shells were prepared through selective  
84 alkylphosphonic acid modification of the Al<sub>2</sub>O<sub>3</sub> surface of  
85 halloysite inner lumen. The micellar features of modified  
86 halloysite are associated with its unique architecture, in which  
87 the hydrophobic lumen allows encapsulation of neutral and  
88 hydrophobic guest molecules by partitioning from a polar  
89 solvent, while the solid polar shell provides stability of the  
90 nanotube dispersion in water and retains the guest molecules.  
91 The stability of conventional organic micelles, which are  
92 characterized by a hydrophobic core and hydrophilic palisade  
93 architecture, depends on many parameters, such as solvent  
94 polarity, critical micelle concentration, and temperature. It is a  
95 rather fluid system; cross-linking may increase micelle stability,  
96 but it may also cause safety concerns because many cross-  
97 linkers are toxic. Our tubule clay micelle-like adsorbent offers  
98 additional features, such as chemical and mechanical stability  
99 and the possibility to produce this material in large quantities.  
100 To characterize specific adsorption and release properties of  
101 clay nanotubes with a hydrophobized interior, we used  
102 hydrophobic and hydrophilic ferrocene derivatives which  
103 were loaded into the halloysite.

104 Previously, ferrocene derivatives have been included in long-  
105 chain alkanes to study the electron-transfer reactions of redox  
106 species in a nonpolar environment.<sup>14</sup> Ferrocenyl and multi-  
107 ferrocenyl systems have been used as redox sensors for  
108 molecular recognition, as building blocks in polymers, and as  
109 electron donors for push–pull nonlinear optical chromophores.  
110 Ferrocene shows biological activity and has been used in  
111 hematinic, antimicrobial, antitumor, and anticancer agents for  
112 potential cancer treatments.<sup>15</sup>

113 To develop selective surface modification of clay nanotubes,  
114 we demonstrated grafting different organic groups to mixed  
115 oxide surfaces of halloysite through successive reactions with  
116 phosphonic acid (interior) and silylating (exterior) agents. Such  
117 a selective bifunctionalization of halloysite nanotubes allows  
118 production of a new type of adsorbent with tunable properties.

## 119 ■ EXPERIMENTAL SECTION

120 **Materials.** Processed halloysite was provided by Applied Minerals  
121 Inc., USA, and used without further treatment. Octadecylphosphonic  
122 acid (ODP), ferrocene (FC), and ferrocenecarboxylic acid (FCA)  
123 were purchased from Aldrich Chemicals and used without further  
124 purification. Tetrahydrofuran (THF) and ethanol were also obtained

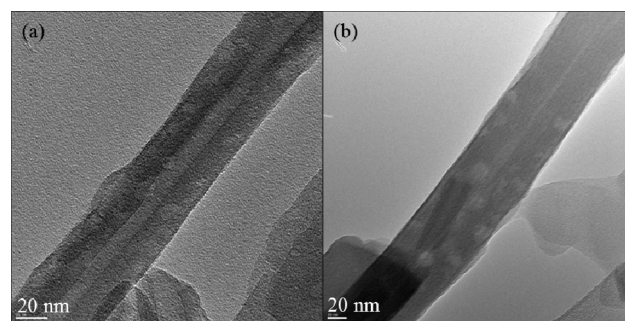
from Aldrich Chemicals. *N*-(2-Aminoethyl)-3-aminopropyltrimethox-  
ysilane (AEAPS) was obtained from Shin-Etsu Chemical Co. Ltd. and  
used as received.

**Modification of Halloysite Lumen with Octadecylphos-  
phonic Acid.** Halloysite (500 mg) was added under stirring to a  
solution of the octadecylphosphonic acid (2 mmol) in 500 mL of 4:1  
EtOH:H<sub>2</sub>O. The EtOH:H<sub>2</sub>O solution was adjusted to pH 4. The  
halloysite suspension was transferred to a vacuum jar, which was then  
evacuated using a vacuum pump. The fizzing of the suspension  
indicated that air has been removed from the lumen of the halloysite  
tubules and replaced with ODP solution.<sup>3</sup> The process of air  
evacuation and cycling back to atmospheric pressure was repeated  
three times in order to maximize ODP in the halloysite lumen. After  
stirring for a week at room temperature, the modified halloysite was  
rinsed, centrifuged five times with EtOH:H<sub>2</sub>O, and dried at 100 °C  
overnight under vacuum. The final product was denoted as halloysite-  
ODP. Halloysite-ODP aqueous suspension showed colloidal stability  
similar to that of untreated halloysite, indicating that the outermost  
surface of the clay nanotube was not hydrophobized.

**Grafting of Organosilane to External Surface of Halloysite-  
ODP.** Further modification of halloysite-ODP by AEAPS was  
performed according to a previously reported procedure.<sup>16b</sup> First,  
0.3 mL of AEAPS dissolved in 80 mL of dry toluene was combined  
with 1 g of dried halloysite-ODP in a glovebox. The suspension was  
dispersed ultrasonically for 20 min at room temperature and heated at  
75 °C under constant stirring for 20 h. The powder was filtered, rinsed  
with 300 mL of toluene to remove excess AEAPS, and then dried at  
100 °C under vacuum.

## 125 ■ RESULTS AND DISCUSSION

**Structural and Morphological Characterization.** Trans-  
mission electron microscopy (TEM) images reveal that  
halloysite is a cylindrical-shaped tube with multilayer walls  
and an open-ended lumen along the nanotube. The samples  
contain agglomerates of nanotubes with some irregularity in  
diameter, wall thickness, and morphology. The tubes have an  
external diameter of 40–60 nm and an inner diameter of 15–  
20 nm, while the wall thickness is about 20 nm. Overall, the  
morphology parameters of our halloysite sample are identical to  
those of previously reported samples.<sup>1,3</sup> After modification of  
halloysite, the transparent central channel (Figure 2a) that runs



**Figure 2.** TEM images of (a) original halloysite and (b) halloysite-ODP.

longitudinally along the nanotube becomes less resolved  
(Figure 2b), indicating that the lumen has been covered with  
a less dense layer, like an organic material. Later, with  
spectroscopic data, we will prove that the layer is bound  
ODP. Judging from the TEM images, the lumen is not  
completely filled or blocked, which may be beneficial for  
loading/release applications.

The porosity of halloysite before and after ODP modification  
was measured by N<sub>2</sub> adsorption–desorption at 77 K (Figure 173

174 S1, Supporting Information). The isotherm of halloysite and  
 175 halloysite-ODP is of type II with H3 hysteresis loops, according  
 176 to IUPAC classification.<sup>16</sup> The BET specific surface area ( $S_{\text{BET}}$ )  
 177 and pore volume ( $V_{\text{pore}}$ ) of untreated halloysite are 43 cm<sup>2</sup>/g  
 178 and 0.53 cm<sup>3</sup>/g, respectively. The isotherm of halloysite-ODP  
 179 is similar to that of original halloysite, with lower  $S_{\text{BET}}$  and  $V_{\text{pore}}$   
 180 of 27.6 cm<sup>2</sup>/g and 0.38 cm<sup>3</sup>/g, respectively. The pore  
 181 distribution curve shows one primary population of pores at  
 182  $32 \pm 1$  and  $28 \pm 1$  nm for halloysite and halloysite-ODP,  
 183 respectively (Table S1, Supporting Information). A 4 nm  
 184 reduction in pore diameter of halloysite after modification  
 185 indicates the formation of a single grafted ODP layer.<sup>17</sup>  
 186 Although the values calculated by N<sub>2</sub> adsorption are biased  
 187 toward maximum diameter compared to values observed by  
 188 TEM images, it is clear that these values have more statistical  
 189 significance for larger size samples.

190 **X-ray Diffraction.** Original halloysite exhibits (001)  
 191 diffraction peak at  $q = 8.33 \text{ nm}^{-1}$ , corresponding to a multilayer  
 192 wall spacing of 0.72 nm, which identifies it as halloysite-7 Å<sup>18</sup>  
 193 (Figure S2, Supporting Information). The diffraction patterns  
 194 in Figure S2 show that the spacing of modified halloysite-ODP  
 195 remained unchanged, indicating that no intercalation of ODP  
 196 into the interlayer of tube walls occurs. This suggests that most  
 197 of the wall interlayer AOH groups of halloysite are unavailable  
 198 for grafting.

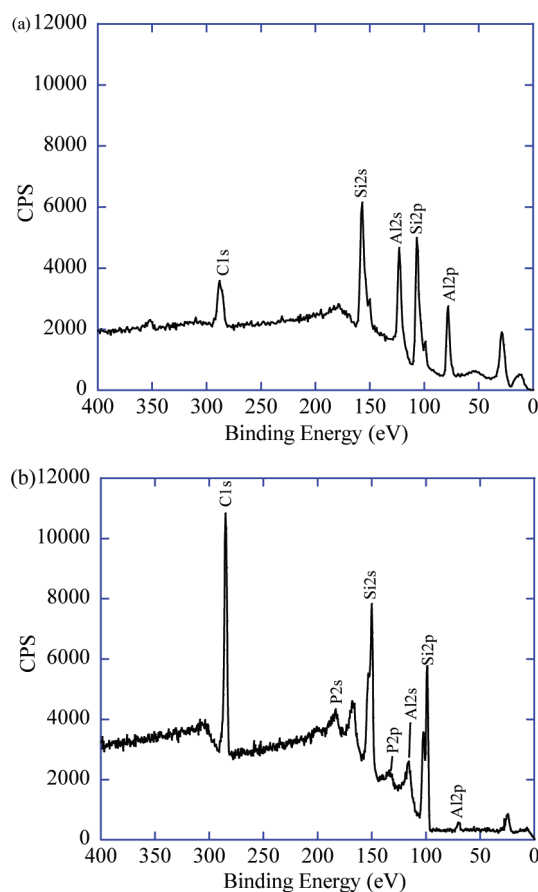
199 **Spectroscopic Data on ODP Immobilization.** *X-ray*  
 200 *Photoelectron Spectroscopy (XPS).* The wall thickness is  
 201 estimated as 20 nm, which well exceeds the penetration depth  
 202 of XPS (10 nm). Thus, it is difficult to disclose the chemical  
 203 composition of the inner lumen surface where the bonding of  
 204 ODP occurs. Nevertheless, since halloysite is oriented  
 205 randomly on a substrate, the inner surface may be partially  
 206 exposed to X-rays. Each silicon atom in halloysite is  
 207 coordinated to four oxygen atoms, and each aluminum atom  
 208 is coordinated to two hydroxyl groups and one oxygen atom  
 209 (Figure 1).<sup>19</sup> Surface chemical compositions of the samples  
 210 were determined (Table 1). Oxygen, silicon, and aluminum are

**Table 1. Atomic Percent Concentration (%) for Elements O 1s, C 1s, P 2p, Si 2p, and Al 2p of Halloysite and Halloysite-ODP**

	O	Si	Al	C	P
original halloysite	67.0	16.5	9.3	7.2	<0.1
halloysite-ODP	34.6	28.7	1.3	34.1	1.7

211 the main elements detected, with a Si/Al ratio of about 1.75,  
 212 higher than unity, indicating the presence of a surface rich in  
 213 silicon. The results show that original halloysite had a small  
 214 amount of carbon (Figure 3a), and that carbon contamination  
 215 likely occurred during sample preparation in the presence of air.

216 After adsorption of ODP (Figure 3b), the intensity of C 1s at  
 217 a binding energy of 286.8 eV increased significantly, with  
 218 detection of a weak P 2p peak at a binding energy of 137 eV, a  
 219 key indication of the presence of phosphonate on halloysite.  
 220 Compared with the theoretical value calculated from the  
 221 stoichiometry of C<sub>18</sub>H<sub>39</sub>O<sub>3</sub>P molecules, the C/P elemental ratio  
 222 provides a check on the chemical composition of the grafted  
 223 layer. The elemental ratio of C/P for halloysite-ODP is 20,  
 224 approaching the theoretical value of 18 with the deviation  
 225 attributed to the hydrocarbon contamination. The main Si 2p  
 226 peak at 107 eV is attributed to Si—O—Si/Si—O—Al.<sup>20</sup> The Si 2p  
 227 of halloysite-ODP is similar to that of the original nontreated



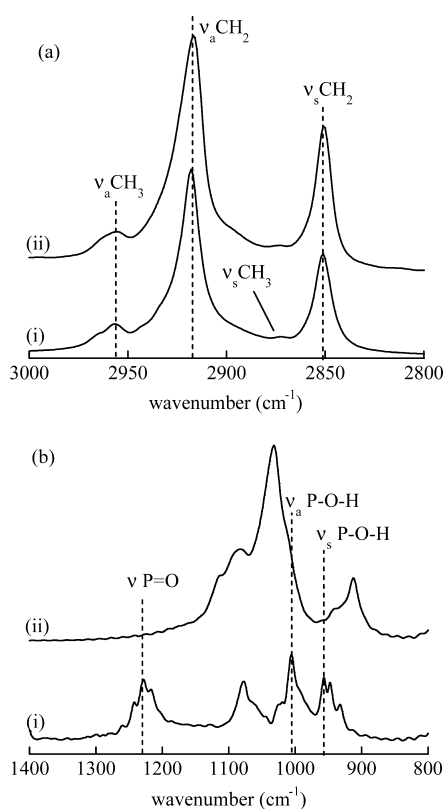
**Figure 3.** XPS spectra taken for (a) unmodified halloysite and (b) halloysite-ODP.

halloysite. No additional component can be assigned to a Si—  
 O—P bond at a binding energy of 103 eV, as reported in the  
 literature.<sup>21</sup> It can be concluded that no Si—O—P formation  
 occurred on the outer surface of halloysite.

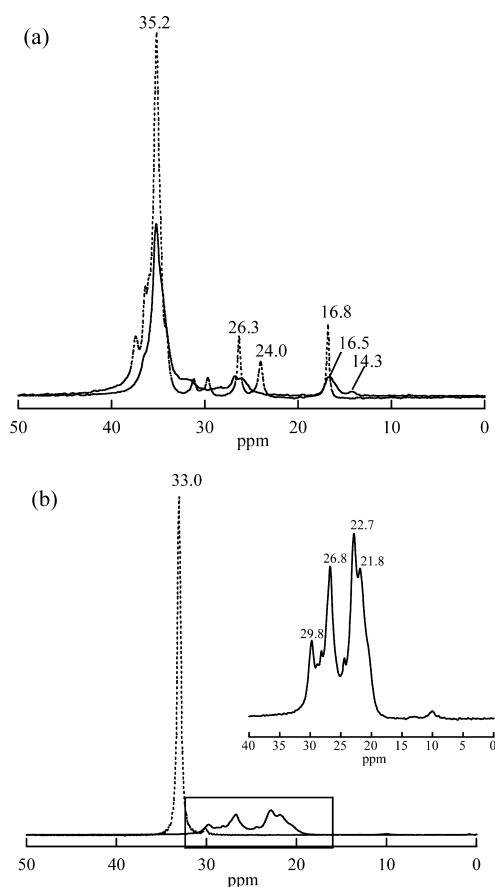
**FTIR Spectroscopy.** Figure 4a shows the methylene  
 stretching region of ODP and halloysite-ODP. The methylene  
 groups are known to give two bands corresponding to  
 symmetric and asymmetric C—H stretching at 2850–2852  
 and 2916–2924 cm<sup>-1</sup>, respectively. In the same region, terminal  
 methyl groups of symmetric and asymmetric stretching modes  
 are detected at 2872 and 2958 cm<sup>-1</sup>. All three frequencies of  
 ODP are shifted toward lower values, which indicates a more  
 constricted vibration of alkyl groups inside the confined spaces  
 of halloysite lumen (Supporting Information, Figure S4, Table  
 S3; the frequency and assignment of each vibrational mode  
 observed are also listed in the Supporting Information).

The dramatic decreases of the P—O—H peak at 2375–2200  
 cm<sup>-1</sup> and the P=O vibration at 1228 cm<sup>-1</sup> could indicate that  
 the binding mode is mostly tridentate (Figure 4b). Definite  
 assignments of these bands are difficult since the ranges for the  
 different P—O stretching peaks overlap and depend on the  
 degree of hydrogen bonding.

**<sup>13</sup>C Solid-State NMR.** A difference in the degree of chain  
 order of ODP adsorbed on halloysite internal surface is  
 revealed by the <sup>13</sup>C CPMAS NMR spectra, shown in Figure 5a.  
 The domain of the ordered and disordered chains is resolved in  
 the <sup>13</sup>C CPMAS NMR spectra because the relative populations  
 of trans and gauche conformations influence the <sup>13</sup>C chemical  
 shift of the interior methylene carbon. For ODP in the



**Figure 4.** (a) Methylene stretching region (3000 to 2800  $\text{cm}^{-1}$ ) and (b) P–O stretching region (1400 to 800  $\text{cm}^{-1}$ ) of the FTIR of (i) ODP and (ii) halloysite-ODP.



**Figure 5.** (a)  $^{13}\text{C}$  CPMAS and (b)  $^{31}\text{P}$  CPMAS NMR spectra of ODP (dotted line) and halloysite-ODP (solid line).

257 crystalline state, these carbons resonate at 33–36 ppm for an  
 258 all-trans conformation.<sup>24</sup> An intense peak at 35.2 ppm,  
 259 corresponding to all-trans chains, is observed for ODP  
 260 adsorbed in the lumen. Whereas a single methyl carbon peak  
 261 at 16.8 ppm is present in the  $^{13}\text{C}$  CPMAS NMR spectra of pure  
 262 crystalline ODP, two methyl carbon peaks are observed at 14.3  
 263 and 16.5 ppm for halloysite-ODP. The upfield shift of the  
 264 methyl group suggests conformational disorder at the chain  
 265 terminals as compared to the methyls in ODP solid.<sup>25</sup>

266  $^{31}\text{P}$  Solid-State NMR. The  $^{31}\text{P}$  solid-state NMR spectrum,  
 267 shown in Figure 5b, reveals the nature of the interaction  
 268 between the phosphonic acid headgroup and a metal oxide  
 269 surface.  $^{31}\text{P}$  chemical shifts are sensitive to variations in the O–  
 270 P–O bond angle as well as to the electronegativity of the  
 271 nearest atoms. ODP headgroups may act as mono-, bi-, or  
 272 tridentate ligands on the alumina surface. Significant changes  
 273 are observed for halloysite-ODP, where the phosphorus  
 274 resonance is shifted further upfield and is inhomogeneously  
 275 broadened. The peaks are shifted from those of the pure ODP,  
 276 and an isotropic chemical shift at 33.0 ppm indicates a strong  
 277 interaction with the alumina surface of halloysite lumen. The  
 278 broad distribution of chemical shifts is due to different types of  
 279 surface bonds and bonding sites on the alumina surface.<sup>26</sup>

280 Based on the average upfield shifts of 6.2 and 10.3 ppm from  
 281 free acid, the primary modes of ODP attachment to the  
 282 alumina surface are bi- and tridentate. Similar  $^{31}\text{P}$  resonance  
 283 shifts have been observed in numerous studies where  
 284 phosphonic acids were reacted with titania, alumina,  $\text{SnO}_2$ ,  
 285 and zirconia surfaces.<sup>27</sup>

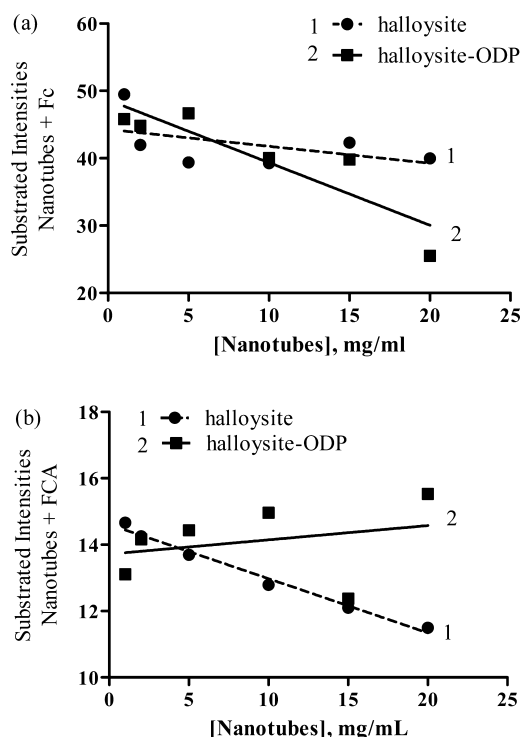
286  $^{29}\text{Si}$  MAS NMR. The  $^{29}\text{Si}$  MAS NMR spectrum of the  
 287 halloysite-ODP (Figure S5, Supporting Information) shows

that there are no high-field resonances arising from higher  
 288 coordinated Si atoms between –120 and –220 ppm. It rules  
 289 out the possibility of phosphoryl groups binding to the silica  
 290 surface, because the presence of phosphonate surface species  
 291 bonded to silica, forming hexa-coordinated Si atoms, would  
 292 give a chemical shift at –212 ppm.<sup>28</sup>

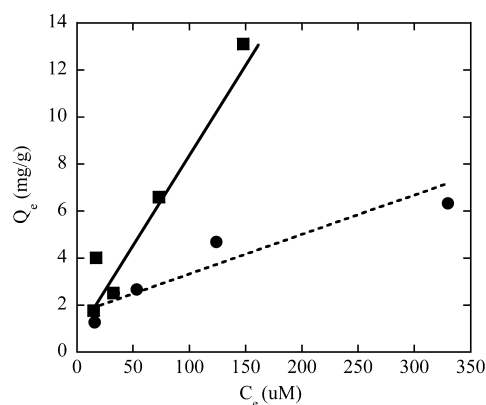
293 Therefore, the following spectroscopic data indicate the  
 294 formation of binding between phosphonic acid and the alumina  
 295 internal surface of halloysite tubes: (1) the absence of a Si–O–  
 296 P bond at the external surface of halloysite from XPS data, (2) a  
 297 significant decrease of P–O–H and P=O vibrations, and (3)  
 298 broadening and upfield shifting in the  $^{31}\text{P}$  CP MAS NMR  
 299 spectrum for halloysite-ODP.

**Adsorption of Ferrocene Derivatives by Halloysite.** To  
 301 compare the adsorption of hydrophobic and hydrophilic  
 302 ferrocene derivatives by original halloysite and halloysite-  
 303 ODP, 1 mM solutions of ferrocene or ferrocenecarboxylic acid  
 304 were added to 5, 10, 15, or 20 mg/mL halloysite samples. After  
 305 2 h of incubation, the samples were centrifuged. The intensities  
 306 of the bands for ferrocene (235–250 nm) and ferrocenecarboxy-  
 307 lic acid (305–320 nm) were obtained for each sample. The  
 308 intensities of the halloysite were subtracted to give the  
 309 absorption of the ferrocene remaining in solution (Figure 6).  
 310 Assuming that ferrocene adsorption is proportional to the  
 311 amount in halloysite samples, linear fits are shown for each  
 312 sample (Table S5, Supporting Information).

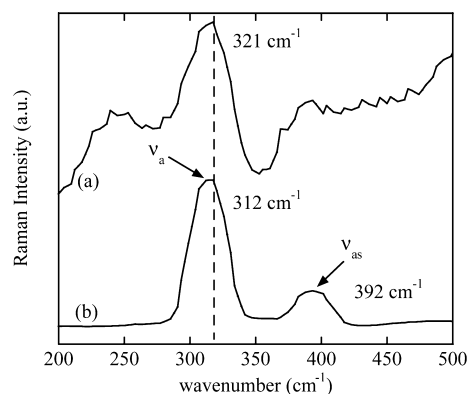
313 The adsorption of ferrocene derivatives is affected by the  
 314 polarity differences of the adsorbent and adsorbate. Ferrocene  
 315 is more hydrophobic, and its adsorption was enhanced by 316



**Figure 6.** Band intensities of absorption of (a) ferrocene and (b) ferrocenecarboxylic acid remaining in solution after halloysite solids were centrifuged out. Various amounts of the halloysites (1, 2, 5, 10, 15, and 20 mg/mL) were added to 1 mM ferrocene/ferrocenecarboxylic acid.



**Figure 7.** Equilibrium adsorption of ferrocene by halloysite (●) and halloysite-ODP (■) at room temperature from THF:H<sub>2</sub>O (1:1) solution.



**Figure 8.** Low-frequency Raman spectra of (a) ferrocene-halloysite-ODP and (b) ferrocene. The ferrocene spectrum is downscaled for better visibility.

317 hydrophobized lumen halloysite-ODP as compared with  
318 untreated halloysite. The halloysite-ODP has higher adsorption  
319 capacity for hydrophobic molecules because the hydrophobic  
320 tubular core acts like a sponge to embed ferrocene into its alkyl  
321 chains. Apparently, modified halloysite-ODP leads to faster  
322 adsorption of ferrocene compared to unmodified halloysite  
323 (Figure S8, Table S6, Supporting Information).

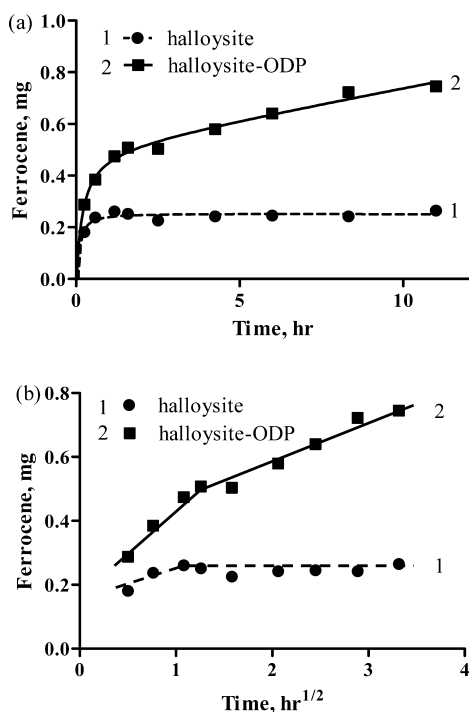
324 For a comparative study, the same experiment and  
325 calculations were performed for adsorption of ferrocenecarbox-  
326 ylic acid. This ferrocene derivative has polar moieties and is  
327 more hydrophilic than ferrocene. We found that untreated  
328 halloysite adsorbed more ferrocenecarboxylic acid than  
329 halloysite-ODP because of the favorable hydrogen bonding in  
330 the alumina lumen. The ODP-modified halloysite showed  
331 barely any adsorption compared to the parent unmodified  
332 halloysite (Figure 6b). The equilibrium adsorption of ferrocene  
333 by halloysite and halloysite-ODP is shown in Figure 7. The  
334 adsorption isotherms for both samples are linear, with  
335 correlation coefficients of 0.88 and 0.95 for halloysite and  
336 halloysite-ODP. The linearity of the adsorption isotherms is  
337 consistent with the idea that inclusion of ferrocene by the  
338 halloysite sample is a partitioning process, in which the  
339 hydrophobic ferrocenes distribute between the organic phase  
340 and aqueous phase. The dominant driving forces for the  
341 ferrocene adsorption are hydrophobic interaction and capillary  
342 forces, where hydrophobic guests are driven into the less polar  
343 lumen of halloysite-ODP. The slope difference between the  
344 isotherms shows that the partitioning equilibria favor the less  
345 polar environment of halloysite-ODP for hydrophobic  
346 ferrocene.

347 **Raman Spectroscopy.** Figure 8 depicts the low-frequency  
348 Raman spectra of ferrocene and ferrocene-halloysite-ODP

349 hybrid. The Raman spectrum of ferrocene within the modified  
350 halloysite-ODP is similar to that reported for ferrocene in the  
351 solution or solid state.<sup>29</sup> The low-frequency Raman spectrum of  
352 ferrocene solid shows two bands at 312 and 392 cm<sup>-1</sup>, which  
353 correspond to the Fe–Cp stretching and Fe–Cp ring tilt,  
354 respectively.<sup>15a</sup> For ferrocene-halloysite-ODP, the Raman signal  
355 is weaker than for the ferrocene powder because of shielding  
356 from the tube wall. We analyzed the strongest peak, at 312  
357 cm<sup>-1</sup>, in the low-frequency spectral region to confirm the  
358 ferrocene inclusion. The Fe–Cp stretch of ferrocene-halloysite-  
359 ODP at 321 cm<sup>-1</sup> is red-shifted in comparison to its position in  
360 ferrocene solid, signifying the elongation of the Fe–Cp distance  
361 for the ferrocene molecule when embedded into the alkyl  
362 chains of halloysite-ODP lumen. The band is also broadened,  
363 which indicates that interactions are present between ferrocene  
364 and alkyl chains.

365 **Release of Ferrocene from Halloysite.** Figure 9a shows  
366 the mass of ferrocene released from samples saturated with 20  
367 mM ferrocene solution at room temperature. For original  
368 halloysite, only a small initial burst release is observed due to  
369 dissolution of the ferrocene from the halloysite surface. ODP-  
370 modified halloysite releases 4 times more ferrocene in the same  
371 period of time, indicating its higher adsorption capacity.

372 The release profiles of ferrocene correspond to diffusional  
373 release and were fitted with a non-Fickian transport model  
374 represented by the Higuchi equation.<sup>30</sup> This model suggests a  
375 dependence of guest release on the square root of time:  $Q_t = 375$



**Figure 9.** (a) Release profile and (b) Higuchi square root of time plots for release of ferrocene from halloysite (●) and halloysite-ODP (■).

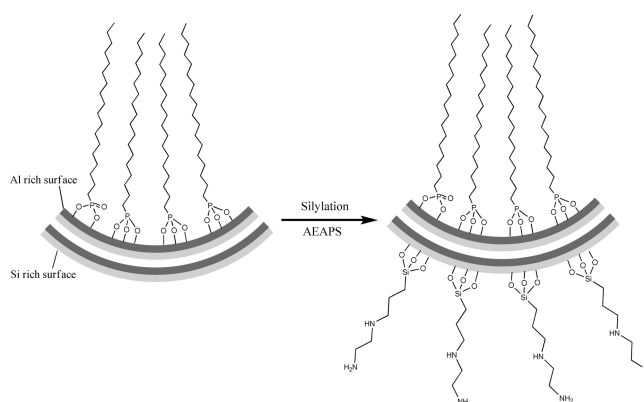
376  $k_H t^{1/2}$ , where  $k_H$  is the release rate. This equation has been  
 377 applied for drug release from porous silica systems.<sup>31</sup> A straight  
 378 line in the plot corresponds to pure Higuchi type of diffusion-  
 379 driven release where no alteration of the matrix occurs.

380 The Higuchi plot for halloysite-ODP displays a two-step  
 381 release (Figure 9b), probably reflecting desorption of the  
 382 outside surface and from the tube lumen. For original  
 383 halloysite, the first linear region is followed by a horizontal  
 384 line, indicating no release at the second stage. The ferrocene  
 385 release rate is higher for the halloysite-ODP system than for the  
 386 original halloysite. Data for unmodified halloysite did not fit  
 387 well to the Higuchi equation, and its release behavior does not  
 388 obey the diffusional release model (Table S7, Supporting  
 389 Information). It is probable that the majority of the ferrocene  
 390 was simply adsorbed on halloysite's external surface.

391 **Bifunctionalization of Halloysite.** After halloysite lumen  
 392 modification through specific reaction of phosphonic acid with  
 393 the alumina innermost surface, we made a consecutive  
 394 treatment of the outermost silica surface through silanization.  
 395 Reaction of halloysite-ODP with AEAPS resulted in the grafting  
 396 of silyl groups, as evidenced by the signal at  $-64.6$  ppm  
 397 characteristic of hydrolyzed AEAPS (Figure S9, Supporting  
 398 Information). Modification of the silica-rich external surface of  
 399 halloysite with silyl groups demonstrates the possibility of  
 400 bifunctionalization of halloysite, as illustrated in Figure 10.

## 401 ■ CONCLUSIONS

402 Selective modification of an aluminosilicate clay nanotube's  
 403 inner lumen with octadecylphosphonic acid was demonstrated.  
 404 In this way we exploited different innermost/outermost surface  
 405 chemistry of clay nanotubes. No octadecylphosphonic acid  
 406 bonding was observed on the siloxane outer tube surface, as  
 407 evidenced by solid-state NMR and XPS. Strong bidentate and  
 408 tridentate Al–O–P bonds were detected on the alumina lumen  
 409 surface of halloysite, since the weakly bound monodentate



**Figure 10.** Schematic illustration of bifunctionalization of the silica–alumina oxide surface of halloysite by ODP and subsequent AEAPS silylation.

410 compounds were removed by sonication and rinsing. The  
 411 adsorption study showed that halloysite with hydrophobic-  
 412 modified lumen adsorbs more ferrocene than its hydrophilic  
 413 derivative (ferrocenecarboxylic acid). Therefore, like in organic  
 414 micelles, the octadecylphosphonic acid immobilized in the  
 415 halloysite lumen may behave as a sponge for physisorption,  
 416 increasing the adsorption capacity for hydrophobic molecules.  
 417 The equilibrium uptake isotherm for ferrocene is linear,  
 418 indicating that these molecules are driven into the lumen by  
 419 partitioning from polar solvent. The release studies also showed  
 420 that halloysite-ODP has higher ferrocene loading. Sequential  
 421 treatment of halloysite with organosilane coupling agents offers  
 422 a simple way to bind covalently a second class of organic groups  
 423 on the silica outermost surface of these clay nanotubes.

## 424 ■ ASSOCIATED CONTENT

### 425 ⓘ Supporting Information

426 Experimental details; figures and tables showing N<sub>2</sub> adsorp-  
 427 tion–desorption isotherm, pore size distribution, wide-angle X-  
 428 ray diffraction pattern, FTIR, <sup>29</sup>Si MAS NMR, UV–visible  
 429 spectra of halloysite before and after modification, adsorption  
 430 results of ferrocene and ferrocenecarboxylic acid, time depend-  
 431 ence of adsorption, release rate of ferrocene by Higuchi  
 432 equation, and <sup>29</sup>Si MAS NMR and <sup>31</sup>P CPMAS NMR after  
 433 consecutive silanization. This material is available free of charge  
 434 via the Internet at <http://pubs.acs.org>.

## 435 ■ AUTHOR INFORMATION

### 436 Corresponding Author

437 takahara@cstf.kyushu-u.ac.jp; ylvov@latech.edu

## 438 ■ ACKNOWLEDGMENTS

439 The authors acknowledge the financial support of a Grant-in-  
 440 Aid for Scientific Research (A) (No. 19205031) from Japan  
 441 Society for the Promotion of Science. The present work is also  
 442 supported by a Grant-in-Aid for GCOE Program, “Science for  
 443 Future Molecular Systems”, from Ministry of Education,  
 444 Culture, Science, Sports and Technology of Japan. Partial  
 445 support by U.S. NSF-1029147 and EPS1003897 grants are  
 446 acknowledged. The synchrotron radiation experiments were  
 447 performed at SPring-8 (No. 2010A1454). The authors thank  
 448 Dr. Elshad Abdullayev (LaTech) and Ms. Keiko Ideta of  
 449 Evaluation Center of Materials Properties and Function, 449

450 Kyushu University, for their assistance in the work and Applied  
451 Minerals, Inc. for supplying halloysite.

## 452 ■ REFERENCES

- 453 (1) (a) Barrientos-Ramírez, S.; Ramos-Fernández, E. V.; Silvestre-  
454 Albero, J.; Sepúlveda-Escribano, A.; Pastor-Blas, M. M.; González-  
455 Montiel, A. *Microporous Mesoporous Mater.* **2009**, *120*, 132–140.  
456 (b) Machado, G. S.; Castro, K.; Wypych, F.; Nagasaki, S. *J. Mol. Catal.*  
457 *A: Chem.* **2008**, *283*, 99–107. (c) Shchukin, D. G.; Sukhorukov, G. B.;  
458 Price, R. R.; Lvov, Y. M. *Small* **2005**, *1*, 510–513. (d) Abdullayev, E.;  
459 Sakakibara, K.; Okamoto, K.; Wey, W.; Ariga, K.; Lvov, Y. *ACS Appl.*  
460 *Mater. Interfaces* **2011**, *3*, 4040–4046.  
461 (2) Wan, C.; Li, M.; Bai, X.; Zhang, Y. *J. Phys. Chem. C* **2009**, *113*,  
462 16238–16246.  
463 (3) (a) Shamsi, M. H.; Geckeler, K. E. *Nanotechnology* **2008**, *19*,  
464 075604–075608. (b) Veerabadran, N. G.; Mongayt, D.; Torchilin, V.;  
465 Price, R. R.; Lvov, Y. *Macromol. Rapid Commun.* **2009**, *30*, 99–103.  
466 (c) Abdullayev, E.; Price, R.; Shchukin, S.; Lvov, Y. *ACS Appl. Mater.*  
467 *Interfaces* **2009**, *1*, 1437–1443. (d) Lvov, Y.; Shchukin, D.; Möhwald,  
468 H.; Price, R. *ACS Nano* **2008**, *2*, 814–820.  
469 (4) (a) Cavallaro, G.; Lazzara, G.; Milioto, S. *Langmuir* **2011**, *27*,  
470 1158–1167. (b) Du, M.; Guo, B.; Jia, D. *Polym. Int.* **2010**, *59*, 574–  
471 582. (c) Wei, W.; Abdullayev, E.; Hollister, A.; Mills, D.; Lvov, Y.  
472 *Macromol. Mater. Eng.* **2012**, *302*, 342–353. (d) Churchman, G. J.;  
473 Carr, R. M. *Clays Clay Miner.* **1975**, *23*, 382–388. (e) Alexander, L. T.;  
474 Faust, G. T.; Hendrick, S. B.; Insley, H.; McMurdie, H. F. *Am. Mineral.*  
475 **1943**, *28*, 1–18. (f) Joussein, E.; Petit, S.; Churchman, J.; Theng, B.;  
476 Righi, D.; Delvaux, B. *Clay Miner.* **2005**, *40*, 383–426. (g) Cavallaro,  
477 G.; Donato, D.; Lazzara, G.; Milioto, S. *J. Phys. Chem. C* **2011**, *115*,  
478 20491–20498.  
479 (5) (a) Hendricks, S. B. *Am. Mineral.* **1938**, *23*, 295–301. (b) Bates,  
480 T. F.; Hildebrand, F. A.; Swineford, A. *Am. Mineral.* **1950**, *35*, 463–  
481 484.  
482 (6) (a) Wada, K. *Am. Mineral.* **1969**, *54*, 50–71. (b) Cradwick, P. D.  
483 *G. Nature* **1972**, *240*, 187–189. (c) Yamamoto, K.; Otsuka, H.; Wada,  
484 S.; Takahara, A. *Chem. Lett.* **2001**, *30*, 1162–1163.  
485 (7) (a) Vergaro, V.; Abdullayev, E.; Lvov, Y. M.; Zeitoun, A.;  
486 Cingolani, R.; Rinaldi, R.; Leporatti, S. *Biomacromolecules* **2010**, *11*,  
487 820–826. (b) Price, R. R.; Gaber, B. P.; Lvov, Y. M. *J.*  
488 *Microencapsulation* **2001**, *18*, 713–722. (c) Veerabadran, N.; Price,  
489 R.; Lvov, Y. *Nano* **2007**, *2*, 215–222.  
490 (8) (a) Hughes, A. D.; King, M. R. *Langmuir* **2010**, *26*, 12155–  
491 12164. (b) Levis, S. R.; Deasy, P. B. *Int. J. Pharm.* **2002**, *243*, 125–134.  
492 (9) (a) Levis, R.; Deasy, P. B. *Int. J. Pharm.* **2003**, *253*, 145–157.  
493 (b) Kelly, H. M.; Deasy, P. B.; Ziaka, E.; Claffey, N. *Int. J. Pharm.*  
494 **2004**, *274*, 167–183.  
495 (10) Hassan-Nejad, M.; Ganster, J.; Bohn, A.; Pinnow, M.; Volkert,  
496 B. *Macromol. Symp.* **2009**, *280*, 123–129.  
497 (11) (a) Laibinis, P. E.; Hickman, J. J.; Wrighton, M. S.; Whitesides,  
498 G. M. *Science* **1989**, *245*, 845–847. (b) Gardner, T. J.; Frisbie, C. D.;  
499 Wrighton, M. S. *J. Am. Chem. Soc.* **1995**, *117*, 6927–6933. (c) Kang,  
500 D.-Y.; Zang, J.; Jones, C. W.; Nair, S. *J. Phys. Chem. C* **2011**, *115*,  
501 7676–7685.  
502 (12) (a) Xia, Y.; Whitesides, G. M. *Adv. Mater.* **1996**, *8*, 765–768.  
503 (b) Ho, P. K. H.; Filas, R. W.; Abusch-Magder, D.; Bao, Z. *Langmuir*  
504 **2002**, *18*, 9625–9628. (c) Sugimura, H.; Nakagiri, N.; Ichinose, N.  
505 *Appl. Phys. Lett.* **1995**, *66*, 3686–3688. (d) Inoue, A.; Ishida, T.; Choi,  
506 N.; Mizutani, W.; Tokumoto, H. *Appl. Phys. Lett.* **1998**, *73*, 1976–  
507 1978.  
508 (13) (a) Johnson, L. M.; Pinnavaia, T. J. *Langmuir* **1990**, *6*, 307–311.  
509 (b) Johnson, L. M.; Pinnavaia, T. J. *Langmuir* **1991**, *7*, 2636–2641.  
510 (14) (a) Michel, R.; Lussi, J. W.; Csucs, G.; Reviakine, I.; Danuser,  
511 G.; Ketterer, B.; Hubbell, J. A.; Textor, M.; Spencer, N. D. *Langmuir*  
512 **2002**, *18*, 3281–3287. (b) Michel, R.; Reviakine, I.; Sutherland, D.;  
513 Fokas, C.; Csucs, G.; Danuser, G.; Spencer, N. D.; Textor, M.  
514 *Langmuir* **2002**, *18*, 8580–8586.  
515 (15) (a) Mohanambe, L.; Vasudevan, S. *Inorg. Chem.* **2005**, *44*,  
516 2128–2130. (b) Podkoscielny, D.; Hooley, R. J.; Rebek, J. Jr.; Kaifer,  
517 A. E. *Org. Lett.* **2008**, *10*, 2865–2868.  
(16) (a) Weber, B.; Serafin, A.; Michie, J.; Rensburg, C. V.; Swarts, J. 518  
C.; Bohm, L. *Anticancer Res.* **2004**, *24*, 763–770. (b) Davis, W.; Shago, 519  
R. F.; Langer, E. H. G.; Swarts, J. C. *Polyhedron* **2005**, *24*, 1611–1616. 520  
(c) Wei, H.; Quan, C. -Y.; Chang, C.; Zhang, X. -Z.; Zhuo, R. -X. *J.* 521  
*Phys. Chem. B* **2010**, *114*, 5309–5314. 522  
(17) (a) Gregg, S. J.; Sing, K. S. W. *Adsorption, Surface Area and* 523  
*Porosity*, 2nd ed: Academic Press: London, 1982. (b) Yuan, P.; 524  
Southon, P. D.; Liu, Z.; Green, M. E. R.; Hook, J. M.; Antill, S. J.; 525  
Kepert, C. J. *J. Phys. Chem. C* **2008**, *112*, 15742–15751. 526  
(18) Ulman, A. *An Introduction to Ultrathin Organic Films: From* 527  
*Langmuir-Blodgett to Self Assembly*; Academic Press: San Diego, CA, 528  
1991. 529  
(19) Brindley, G. W. Order-disorder in the clay mineral structures. In 530  
*Crystal Structures of Clay Minerals and Their X-ray Identification*; 531  
Brindley, G. W., Brown, G., Eds.; Mineralogist Society: London, 1980; 532  
p 125. 533  
(20) (a) Bates, T. F.; Hildebrand, F. A.; Swineford, A. *Am. Mineral.* 534  
**1950**, *35*, 463–484. (b) Singh, B. *Clays Clay Miner.* **1996**, *44*, 191– 535  
196. (c) Singh, B.; Gilkes, R. J. *Clays Clay Miner.* **1992**, *40*, 212–229. 536  
(d) Singh, B.; Mackinnon, I. D. R. *Clays Clay Miner.* **1996**, *44*, 825– 537  
834. 538  
(21) Wittberg, T. N.; Wang, P. S. *Surf. Interface Anal.* **1999**, *27*, 936– 539  
940. 540  
(22) Massiot, Ph.; Centeno, M. A.; Carrizosa, I.; Odriozola, J. A. *J.* 541  
*Non-Cryst. Solids* **2001**, *292*, 158–166. 542  
(23) (a) Gao, W.; Dickinson, L.; Grozinger, C.; Morin, F. G.; Reven, 543  
L. *Langmuir* **1996**, *12*, 6429–6435. (b) Frost, R. L.; Kristof, J. *Clays* 544  
*Clay Miner.* **1997**, *45*, 551–563. (c) Madejova, J.; Komadel, P. *Clays* 545  
*Clay Miner.* **2001**, *49*, 410–432. 546  
(24) (a) Earl, W. L.; VanderHart, D. L. *Macromolecules* **1979**, *12*, 547  
762–767. (b) Tonelli, A. E.; Schilling, F. C. *Acc. Chem. Res.* **1981**, *14*, 548  
233–238. 549  
(25) Pursch, M.; Strohschein, S.; Händel, H.; Albert, K. *Anal. Chem.* 550  
**1996**, *68*, 386–393. 551  
(26) (a) Cao, G.; Hong, H. -G.; Mallouk, T. E. *Acc. Chem. Res.* **1992**, 552  
*25*, 420–427. (b) Lynch, V. M.; Mallouk, T. E. *Inorg. Chem.* **1988**, *27*, 553  
2781–2785. 554  
(27) (a) McNatt, J. S.; Morgan, J. M.; Farkas, N.; Ramsier, R. D. 555  
*Langmuir* **2003**, *19*, 1148–1153. (b) Holland, G. P.; Sharma, R.; 556  
Agola, J. O.; Amin, S.; Solomon, V. C.; Singh, P.; Buttry, D. A.; Yarger, 557  
J. L. *Chem. Mater.* **2007**, *19*, 2519–2526. (c) Lurkes, I.; Borbaruah, M.; 558  
Quin, L. D. *J. Am. Chem. Soc.* **1994**, *116*, 1737–1741. (d) Sasaki, D. Y.; 559  
Alam, T. M. *Chem. Mater.* **2000**, *12*, 1400–1407. (e) Pawsey, S.; 560  
McCormick, M.; De Paul, S.; Graf, R.; Lee, Y. S.; Reven, L.; Spiess, H. 561  
W. *J. Am. Chem. Soc.* **2003**, *125*, 4174–4184. 562  
(28) Corriu, R. J. P.; Leclercq, D.; Mutin, P. H.; Sarlin, L.; Vioux, A. *J.* 563  
*Mater. Chem.* **1998**, *8*, 1827–1833. 564  
(29) Ellis, R. L.; Nelson, R. D. *Spectrochim. Acta* **1958**, *10*, 307–329. 565  
(30) (a) Higuchi, T. *J. Pharm. Sci.* **1961**, *50*, 874–875. (b) Higuchi, 566  
T. *J. Pharm. Sci.* **1963**, *52*, 1145–1149. 567  
(31) (a) Vallet-Regí, M.; Balas, F.; Acros, D. *Angew. Chem., Int. Ed.* 568  
**2007**, *46*, 7548–7558. (b) Munoz, B.; Rámila, A.; Pérez-Pariente, J.; 569  
Díaz, I.; Vallet-Regí, M. *Chem. Mater.* **2003**, *15*, 500–503. 570  
(c) Anderson, J.; Rosenholm, J.; Areva, S.; Linden, M. *Chem. Mater.* 571  
**2004**, *16*, 4160–4164. 572